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Supplementary Materials for
Visible-frequency long-distance subwavelength focusing
via surface plasmon hyperbolic polaritons (SPHPs)

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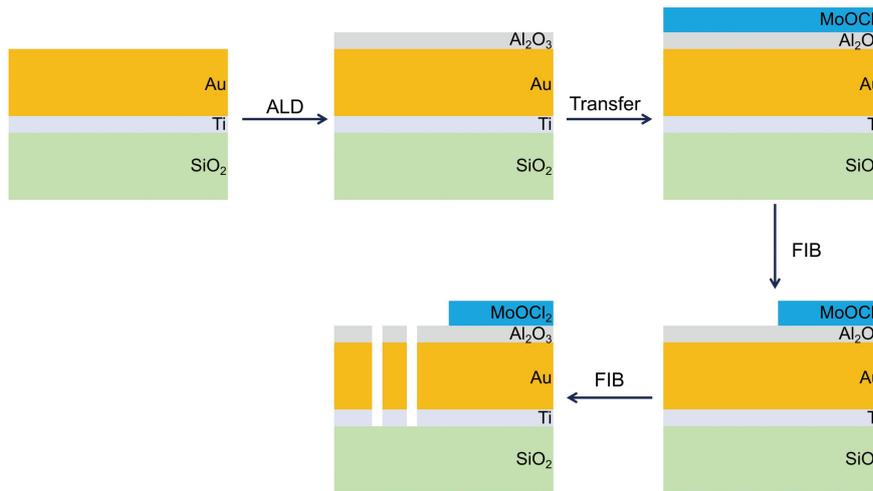
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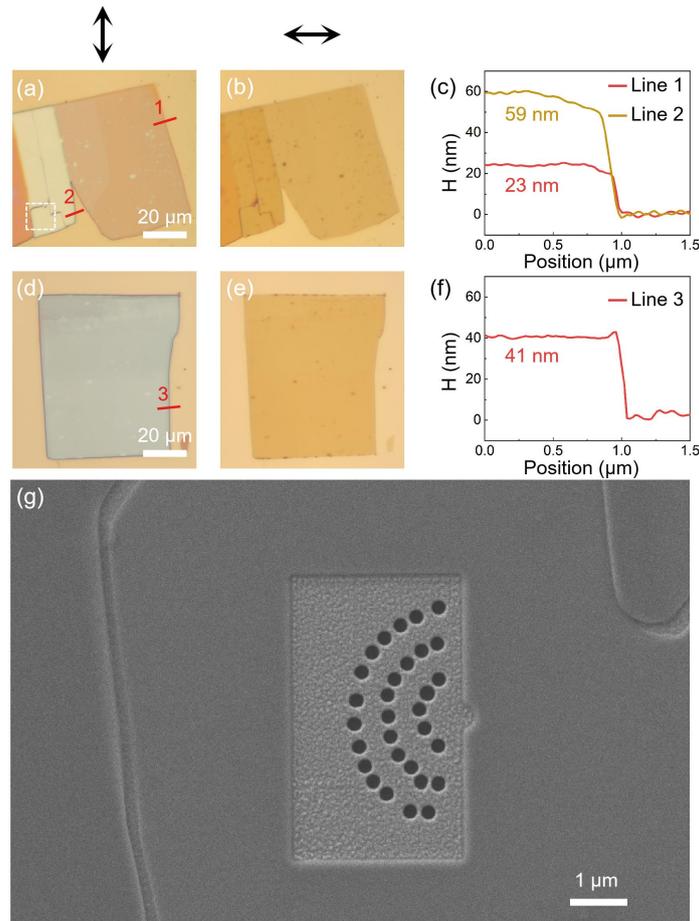
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Figure S1. Sample fabrication process.

21 The fabrication process began with the sequential deposition of a 4 nm Ti adhesion layer and a
 22 110 nm Au layer onto a quartz substrate using electron beam evaporation. Subsequently, a 2 nm
 23 Al₂O₃ layer was grown at 200°C via atomic layer deposition. The MoOCl₂ flakes with designated
 24 thicknesses were then transferred onto the substrate through a dry-transfer process.

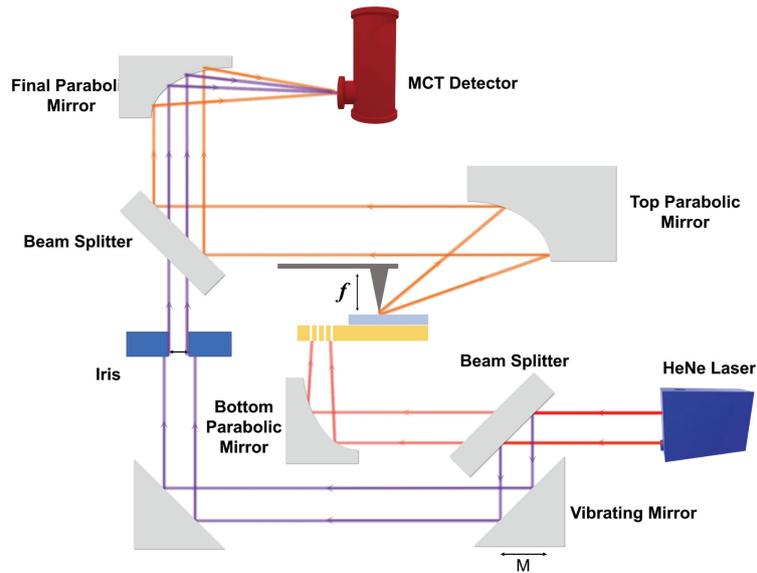
25 To define the target nanostructure, a two-step focused ion beam (FIB) milling procedure was
 26 employed. In the first step, the MoOCl₂ layer was partially milled to expose the underlying Au
 27 film and to create a clean edge profile of the MoOCl₂, leaving a well-defined uncovered Au region
 28 for undisturbed surface plasmon polariton (SPP) propagation. In the second step, through-hole
 29 arrays were patterned into the exposed Au region to complete the nanostructure layout.



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31 **Figure S2.** Sample characterization. (a,d) and (b,e) Optical images of transferred MoOCl₂ flakes
 32 on Au substrate under y- and x-polarized light (arrows indicate polarization direction). (c) and (f)
 33 Corresponding AFM height profiles. (g) Top-view SEM image of the FIB-patterned structure in
 34 the white dashed box in (a).

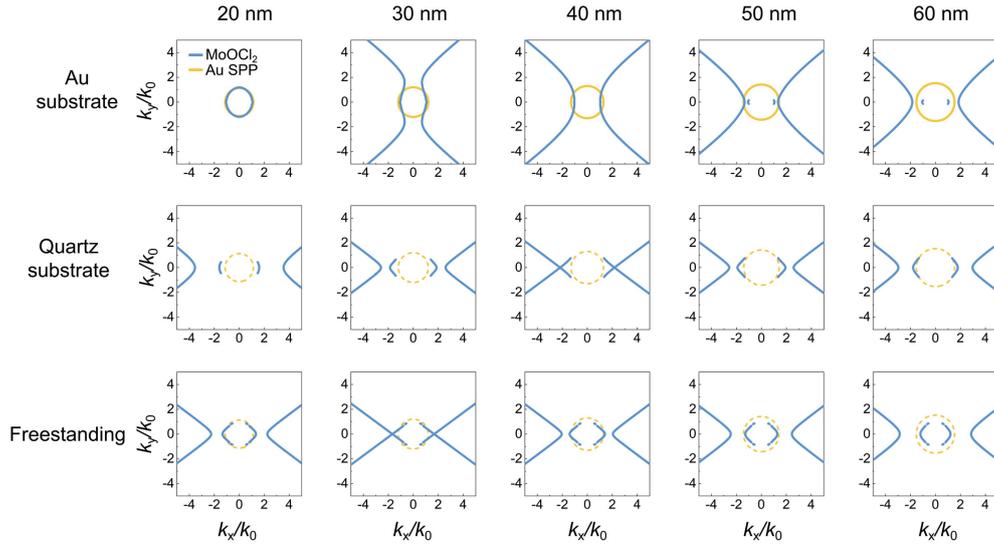
35 Optical images of MoOCl₂ flakes with different thicknesses transferred onto the Au substrate
 36 are shown in Figure S2(a,d) and (b,e) under y- and x-polarized illumination, respectively
 37 (polarization directions indicated by arrows at the top). Thicknesses of the three MoOCl₂ regions
 38 measured by atomic force microscopy (AFM) are shown in Figure S2(c) and (f), respectively.
 39 Figure S2(g) presents a top-view scanning electron microscope (SEM) image of the FIB-patterned
 40 structure fabricated within the white dashed box in Figure S2(a), showing the MoOCl₂ edge and
 41 the 30-hole metastructure on the exposed Au film.



42

43 **Figure S3.** Complete transmission-mode SNOM setup for near-field measurements.

44 The transmission-mode SNOM setup for near-field measurements is shown in Figure S3, where
 45 the laser beam is focused by a bottom parabolic mirror and vertically incident onto the bottom of
 46 the sample, exciting the plasmonic responses of the Au nanostructure. The surface electric field is
 47 scattered by the tip, collected by a top parabolic mirror, and ultimately directed towards a
 48 photodetector for measurement. To improve sensitivity, pseudo-heterodyne demodulation is
 49 implemented via an additional mirror vibration path, allowing extraction of higher-order near-field
 50 signals. This configuration provides amplitude- and phase-resolved mapping of the near-field
 51 distribution while effectively minimizing direct laser excitation and tip-induced field changes, thus
 52 enabling clean SPP propagation and precise wavefront detection.



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Figure S4. Thickness and dielectric environment dependent MoOCl₂ IFCs.

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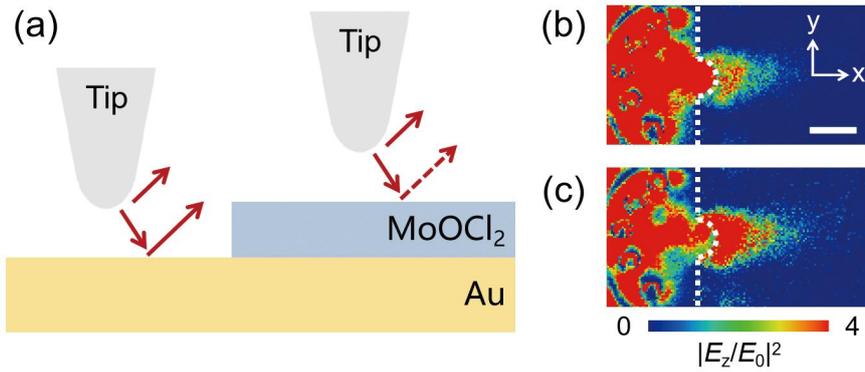
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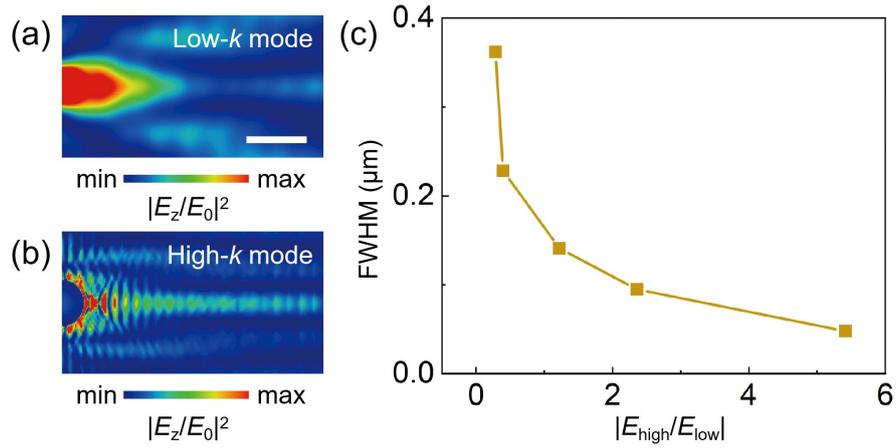
Figure S4 shows the calculated IFCs of MoOCl₂ on Au substrate, quartz substrate and in freestanding form for different thicknesses. Although the dielectric environment modifies the hyperbola shape, the hyperbolic vertices on quartz substrate and in freestanding case remain beyond k_{SPP} , indicating that intrinsic hyperbolic polariton modes are restricted to high- k components and cannot access lower wavevectors. In contrast, MoOCl₂ on the Au substrate supports hyperbolic modes at lower k within an appropriate thickness range, enabling effective wavefront control through the surface plasmon hyperbolic polariton (SPHP) mode.



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63 **Figure S5.** Quantitative correction for intrinsic near-field mapping. (a) Dielectric-induced signal
 64 contrast across different regions under transmission-mode SNOM detection. (b) Raw near-field
 65 electric field distribution measured by SNOM. (c) Intrinsic near-field electric field distribution
 66 after quantitative correction. Scale bar, 500 nm.

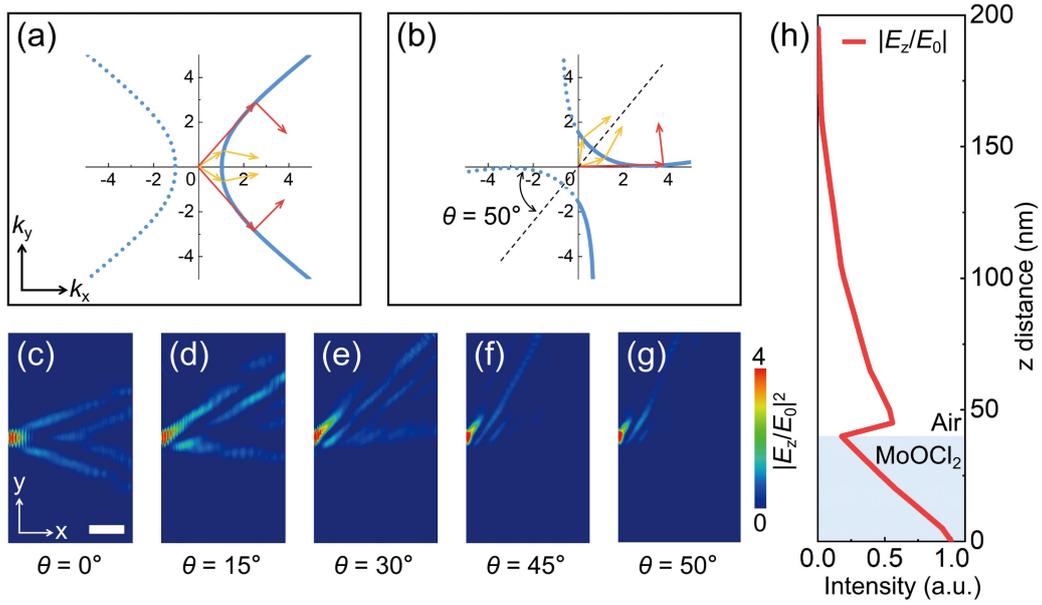
67 The detected SNOM signal contains a dielectric-dependent contribution arising from
 68 surface-reflected scattering, leading to artificial intensity contrast between Au and MoOCl₂
 69 regions, as shown in Figure S5(a). By applying a quantitative correction method established in our
 70 previous work¹, this dielectric-induced disparity is removed, as shown in Figure S5(b) and (c),
 71 enabling reconstruction of the intrinsic near-field electric field distribution and allowing reliable
 72 comparison of absolute field intensities across different material regions.



73

74 **Figure S6.** Momentum-resolved near-field analysis via FFT filtering. (a) and (b) Low-*k* and
 75 high-*k* near-field electric field distributions extracted by two-dimensional FFT filtering for the 40
 76 nm MoOCl₂ sample, respectively. (c) FWHM of the focal spot as a function of the ratio $|E_{\text{high}}/E_{\text{low}}|$
 77 for different MoOCl₂ thicknesses. Scale bar, 500 nm.

78 Two-dimensional fast Fourier transform (FFT) filtering of the simulation results is used to
 79 separate low- and high-momentum components of the surface electric field, as shown in Figure S6.
 80 The low-*k* component supports long-range propagation, while the high-*k* component provides
 81 strong spatial confinement. Their cooperative contribution enables long-distance subwavelength
 82 focusing. The monotonic decrease of the full width at half maximum (FWHM) with increasing
 83 $|E_{\text{high}}/E_{\text{low}}|$ demonstrates the critical role of high-*k* components in narrowing the focal spot.



84

85 **Figure S7.** Separation of low- k and high- k components via crystal-axis rotation. (a) and (b) IFCs
 86 at $\theta = 0^\circ$ and 50° , with solid and dashed curves representing excitable and momentum-mismatched
 87 modes, respectively. (c)-(g) Simulated field intensity distributions for $\theta = 0^\circ, 15^\circ, 30^\circ, 45^\circ, 50^\circ$,
 88 respectively. (h) z -profile of electric field at focus for the $\theta = 50^\circ$ case. Scale bar, 1 μm .

89 Owing to the anisotropic hyperbolic dispersion, changing the angle θ between the crystal axis
 90 and the incidence direction redirects the propagation of low- k and high- k modes, enabling
 91 momentum-selective filtering. At $\theta = 0^\circ$ (Figure S7(a)), both low- k and high- k modes are allowed
 92 and jointly contribute to focusing, whereas at $\theta = 50^\circ$ (Figure S7(b)) the high- k modes are steered
 93 away from the MoOCl₂ region, leaving a pure low- k response. Corresponding simulated field
 94 distributions in Figure S7(c)-(g) show a progressive suppression of high- k contributions with
 95 increasing θ , which vanish above 45° . The z -profile of the focal field at $\theta = 50^\circ$ (Figure S7(h))
 96 confirms that the remaining signal is a hybrid surface mode dominated by metal-supported SPPs,
 97 whose propagation is modulated by the in-plane hyperbolic dispersion of the MoOCl₂ layer.

98 **Reference**

- 99 1. Y. Fan et al., Direct measurement of the real strength of near-field electric field. *Appl. Phys.*
100 *Lett.* **125**, 211105 (2024).