

SUPPLEMENTARY INFORMATION

Direct Imaging of Composition-driven Magnetoelastic Phase Transformations in Bulk Fe–Rh

K. Padrón-Alemán, J. Shen, A. Tengattini, L. Helfen, G.J. Cuello, J.A. Blanco, J.L. Sánchez Llamazares, P. Alvarez-Alonso and Pedro Gorria

Affiliations

Institut Laue-Langevin, Grenoble, France

K. Padrón-Alemán, J. Shen, A. Tengattini, L. Helfen, G.J. Cuello

Departamento de Física, Universidad de Oviedo, Oviedo, Spain

K. Padrón-Alemán, J.A. Blanco, P. Alvarez-Alonso, Pedro Gorria

University Grenoble Alpes, CNRS, Grenoble, France

A. Tengattini

Instituto Potosino de Investigación Científica y Tecnológica A.C., San Luis Potosí, México

J.L. Sánchez Llamazares

Instituto Universitario de Tecnología Industrial de Asturias, Universidad de Oviedo, Gijón, Spain

P. Alvarez-Alonso, Pedro Gorria

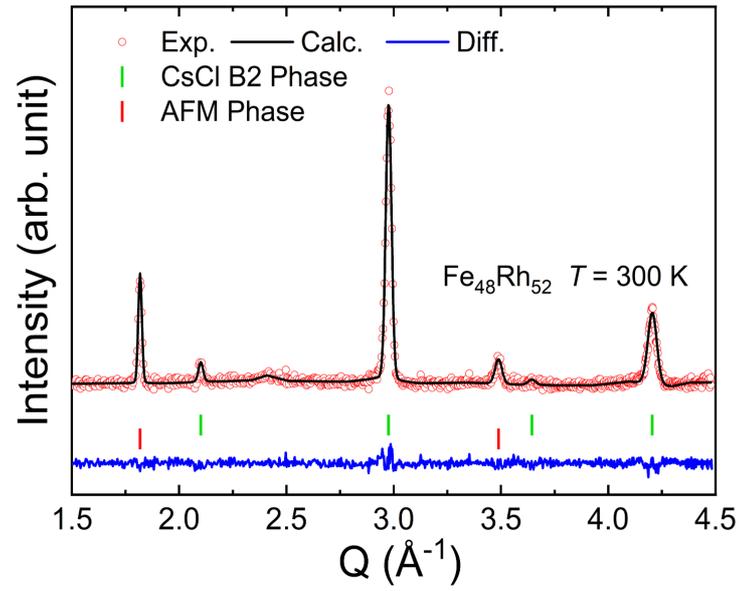


Fig. S1. Crystal and magnetic structures. Neutron diffraction pattern of the $\text{Fe}_{48}\text{Rh}_{52}$ alloy collected at 300 K. Green and red bars indicate the reflections corresponding to the nuclear and magnetic (AFM) structures, respectively.

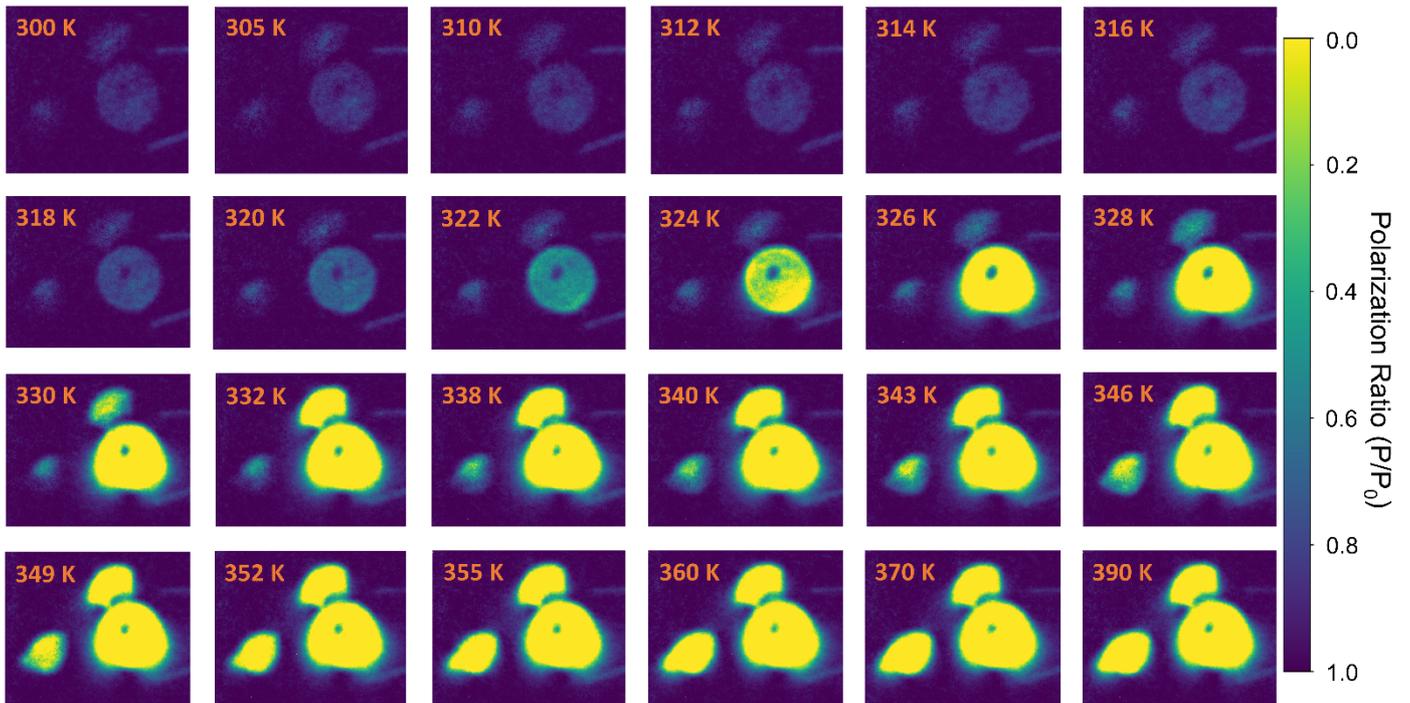


Fig. S2. AFM-FM depolarization-contrast radiographs. Depolarization-contrast radiographs resolving the AFM→FM phase transition in equiatomic and near-equiatomic Fe–Rh alloys, obtained using the polarized neutron imaging technique. Fe₅₀Rh₅₀ alloy is shown on the left, Fe₄₉Rh₅₁ at the top, and Fe₄₈Rh₅₂ at the bottom right.

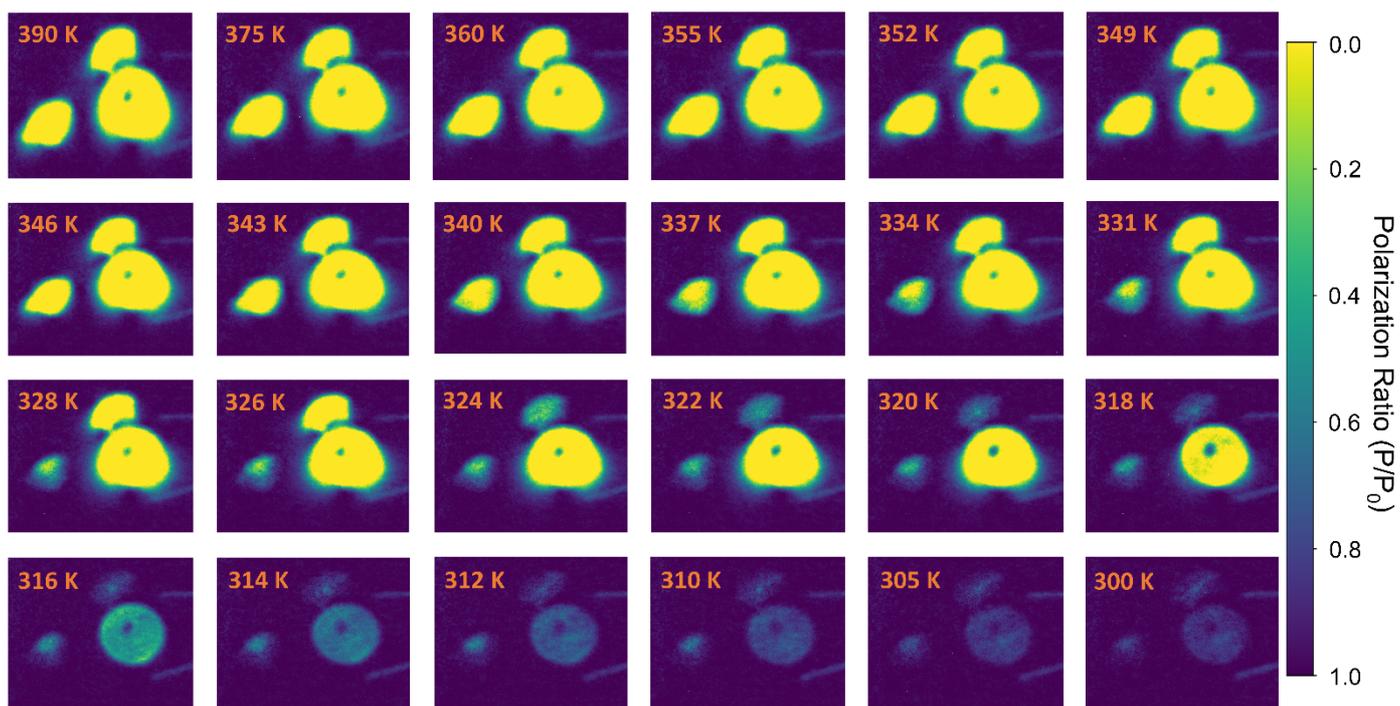


Fig. S3. FM-AFM depolarization-contrast radiographs. Depolarization-contrast radiographs resolving the FM→AFM phase transition in equiatomic and near-equiatomic Fe–Rh alloys, obtained using the polarized neutron imaging technique. $\text{Fe}_{50}\text{Rh}_{50}$ alloy is shown on the left, $\text{Fe}_{49}\text{Rh}_{51}$ at the top, and $\text{Fe}_{48}\text{Rh}_{52}$ at the bottom right.

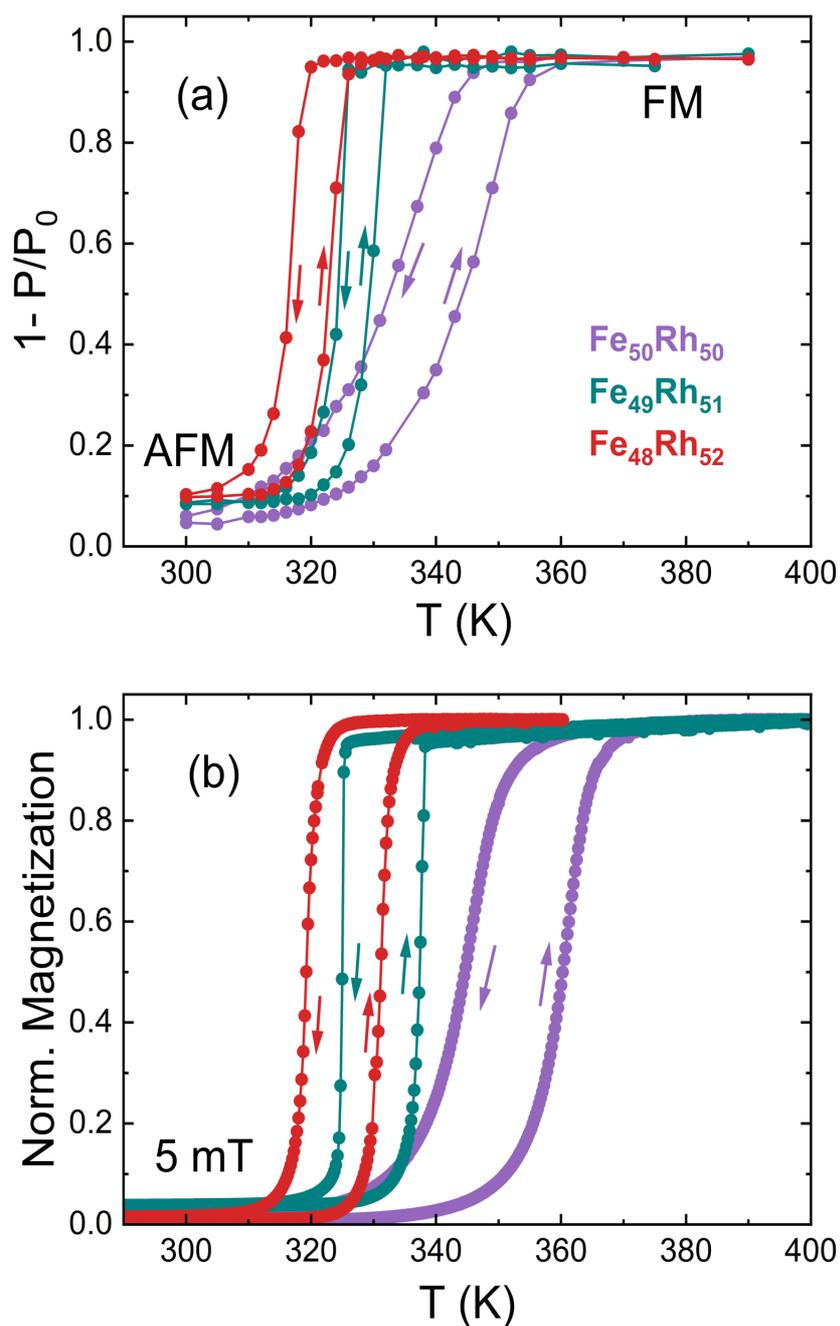


Fig. S4. Temperature-dependent averaged $1-P/P_0$ and M/M^{\max} . **a** Temperature dependence of the averaged $1-P/P_0$ over the full FOV, and **b** normalized 5 mT magnetization curve for $\text{Fe}_{50}\text{Rh}_{50}$, $\text{Fe}_{49}\text{Rh}_{51}$, and $\text{Fe}_{48}\text{Rh}_{52}$ samples. Arrows pointing up and down indicate heating and cooling procedures, respectively.

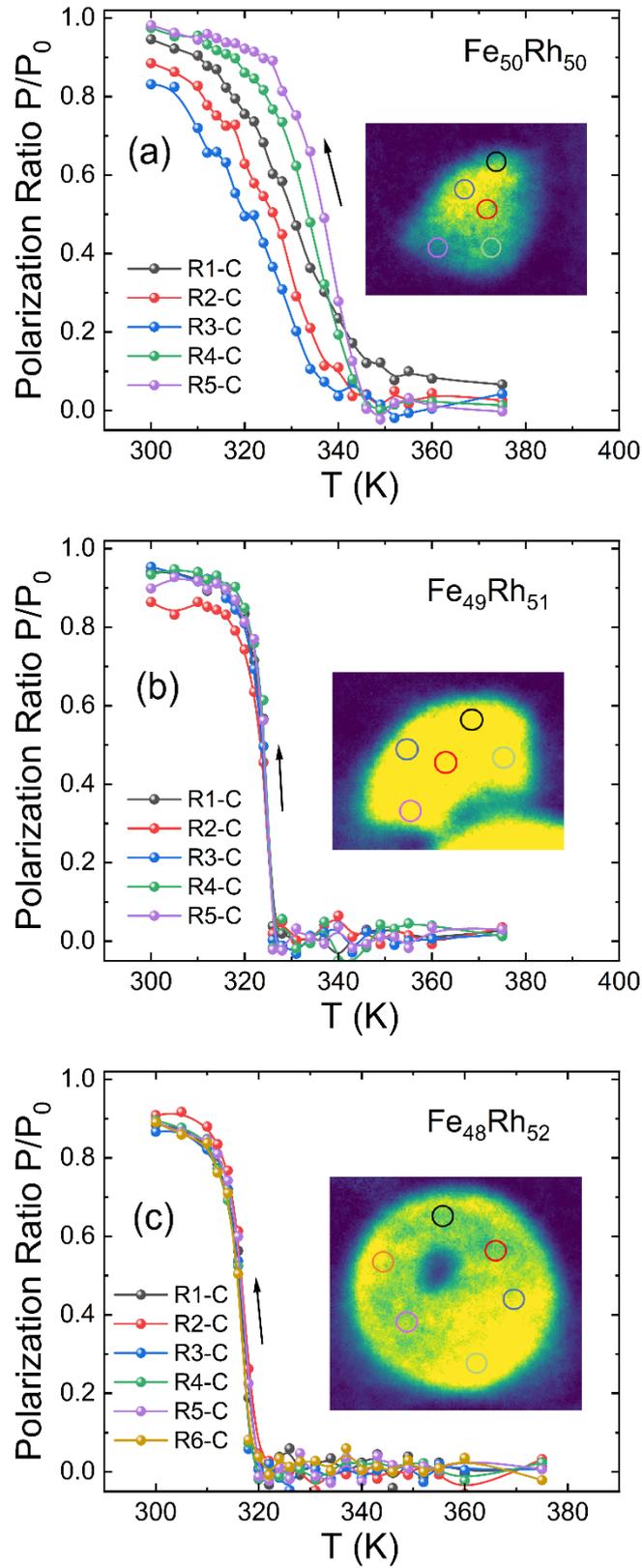


Fig. S5. Temperature-dependent averaged polarization ratio. Evolution of the polarization ratio during cooling process calculated from selected areas within the samples for **a** $\text{Fe}_{50}\text{Rh}_{50}$, **b** $\text{Fe}_{49}\text{Rh}_{51}$, and **c** $\text{Fe}_{48}\text{Rh}_{52}$. The black arrows indicate the temperature-change direction. Each encircled area corresponds to a region of 10 pixels in diameter ($\approx 570 \mu\text{m}$). The accompanying images illustrate the sampled regions (circled).

Video S1. The video shows the sequence of depolarization-contrast radiographs presented in Fig. S3 and Fig. S4 during heating (AFM→FM transition) and cooling (FM→AFM transition), respectively.

Table S1. Relevant Parameters of the Phase Transformation. Transition temperatures obtained from the low-field $M(T)^{5\text{ mT}}$ curves. Start (AF_S , FA_S) and finish (AF_f , FA_f) temperatures determined using the tangent method, thermal hysteresis (ΔT_{hyst}) calculated as $\Delta T_{\text{hyst}} = [AF_S + AF_f - (FA_S + FA_f)]/2$, and ΔM at the AFM↔FM transition for an applied field of 2 T.

Alloy	T_t (K) $M(T)^{5\text{ mT}}$	$\mu_0 H$ (T)	AF_S (K)	AF_f (K)	FA_S (K)	FA_f (K)	ΔT_{hyst} (K)	ΔM (A m ² kg ⁻¹)
Fe ₅₀ Rh ₅₀	352	2	336	369	353	318	17	112
Fe ₄₉ Rh ₅₁	335	2	321	322	309	307	14	123
Fe ₄₈ Rh ₅₂	328	2	313	319	305	300	14	114