

# Supplementary Information

Synthesizer: Machine Learning-Guided Perovskite Nanocrystal Optimization

**Nina A. Henke\*** <sup>1</sup>, **Leo Lubert\*** <sup>1</sup>, **Ioannis Kouroudis\*** <sup>2</sup>, **Jonathan Paul**<sup>1</sup>,  
**Alexander Schuhbeck**<sup>1</sup>, **Lukas M. Rescher**<sup>3</sup>, **Tizian Lorenzen**<sup>4</sup>, **Veronika Mayer**<sup>1</sup>,  
**Knut Müller-Caspary** <sup>4</sup>, **Bert Nickel** <sup>3</sup>, **Alessio Gagliardi** <sup>2</sup>, **Alexander S. Urban** <sup>1</sup> 

<sup>1</sup>Nanospectroscopy Group and Center for NanoScience, Faculty of Physics, Ludwig-Maximilians-University Munich, Königinstraße 10, 80539 Munich, Germany

<sup>2</sup>Chair of Simulation of Nanosystems for Energy Conversion, Department of Electrical Engineering, TUM School of Computation, Information and Technology, Atomistic Modeling Center (AMC), Munich Data Science Institute (MDSI), Technical University of Munich, Hans-Piloty-Straße 1, 85748 Garching, Germany

<sup>3</sup>Soft Condensed Matter Group and Center for NanoScience, Faculty of Physics, Ludwig-Maximilians-University Munich, Geschwister-Scholl-Platz 1, 80539 Munich, Germany

<sup>4</sup>Department of Chemistry and Center for NanoScience, Ludwig-Maximilians-University Munich, Butenandtstraße 11, 81377 Munich, Germany

\* These authors contributed equally: Nina A. Henke, Leo Lubert, Ioannis Kouroudis.

 [urban@lmu.de](mailto:urban@lmu.de)

April 09, 2025

## Materials

Cesium carbonate ( $\text{Cs}_2\text{CO}_3$ , 99%, Sigma Aldrich), lead bromide ( $\text{PbBr}_2$ , 98%, Sigma Aldrich), lead iodide ( $\text{PbI}_2$ , 99%, Alfa Aesar), oleic acid (OAc, technical grade, 90%, Sigma Aldrich), oleylamine (OAm, technical grade, 70%, Sigma Aldrich), toluene (99.9%, VWR Chemicals), n-hexane (97%, VWR Chemicals), methanol (MeOH, 99.8%, Sigma Aldrich), ethanol (EtOH, 99.5%, Sigma Aldrich), isopropanol (i-PrOH, for analysis, Merck), n-butanol (n-BuOH, 99%, Sigma Aldrich), cyclopentanone (CyPen, for synthesis, Merck), methyl acetate (99%, for synthesis, Merck). All chemicals listed above were used as received without further purification.

## Syntheses

**Supplementary Table 1 | Constraints for the synthesis of  $\text{CsPbBr}_3$  nanocrystals.** Summary of experimental constraints for the preparation of perovskite nanocrystals by antisolvent-assisted spontaneous crystallization.

experimental constraint	notes
$\text{OAm/PbBr}_2 = 2.1$	fixed ligand ratio
$\text{OAc/PbBr}_2 = 2.8$	fixed ligand ratio
$0.020 \text{ M} \leq c_{\text{Cs}} \leq 0.200 \text{ M}$	precursor concentration
$0.001 \text{ M} \leq c_{\text{PbBr}_2} \leq 0.100 \text{ M}$	precursor concentration
$50 \mu\text{L} \leq V_{\text{Cs}} \leq 5000 \mu\text{L}$	pipetting
$500 \mu\text{L} \leq V_{\text{PbBr}_2} \leq 5000 \mu\text{L}$	pipetting
$0 \leq V_{\text{as}} \leq 5000 \mu\text{L}$	pipetting
$V_{\text{Cs}} + V_{\text{PbBr}_2} + V_{\text{as}} \leq 10 \text{ mL}$	reaction vial

**Supplementary Table 2 | Classification of CsPbBr<sub>3</sub> nanocrystals.** Perovskite nanocrystal were sorted into different groups according to the PL peak wavelength or PL peak energy. The upper and lower limit for each nanocrystal product is listed.

	lower limit (nm)	upper limit (nm)	upper limit (eV)	lower limit (eV)
2	429	437	2.89	2.84
3	455	465	2.72	2.67
4	470	480	2.63	2.58
5	481	489	2.58	2.54
6	490	497	2.53	2.49
7	498	504	2.49	2.46
8	505	509	2.46	2.44
∞	510	520	2.43	2.38

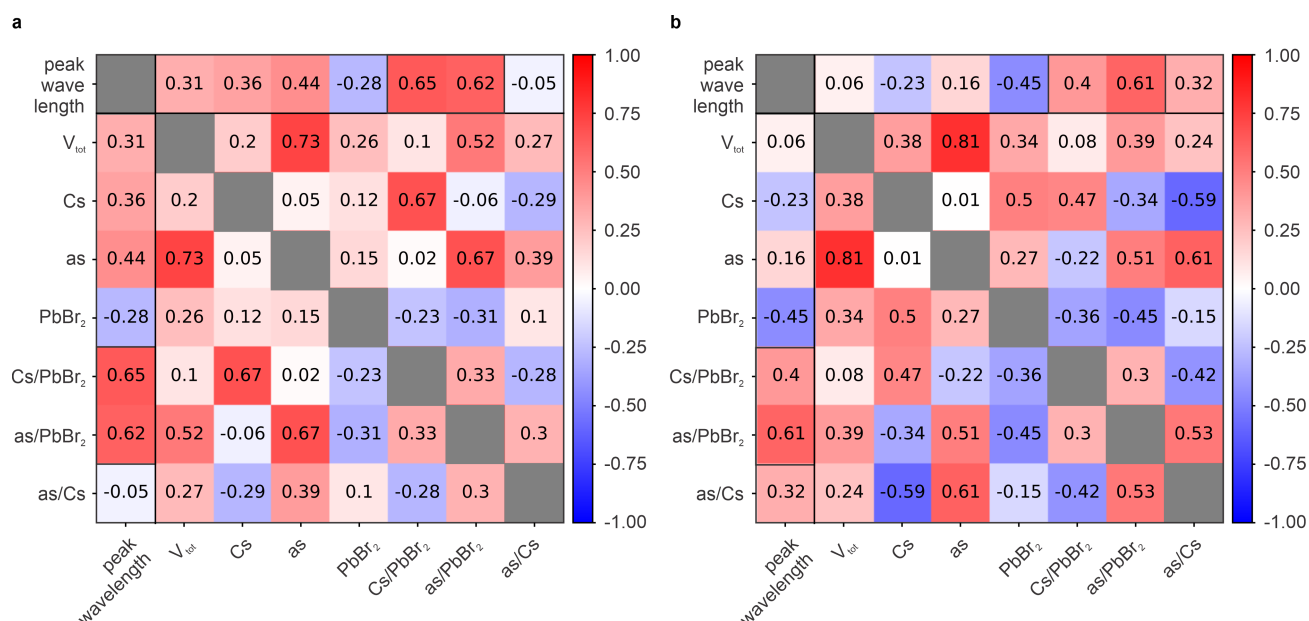
**Supplementary Table 3 | Summary of syntheses.** Nanocrystal syntheses with different antisolvents are listed, including details on number of syntheses with monodisperse, polydisperse and non-emissive samples as products. The label 'polydisperse' is used to describe synthesis products which exhibit multiple emission peaks.

antisolvent	syntheses	monodisperse	polydisperse	non-emissive
MeOH	65	58	6	1
EtOH	71	60	8	3
i-PrOH	33	26	7	0
n-BuOH	45	35	10	0
CyPen	57	56	1	0
Toluene	16	15	1	0

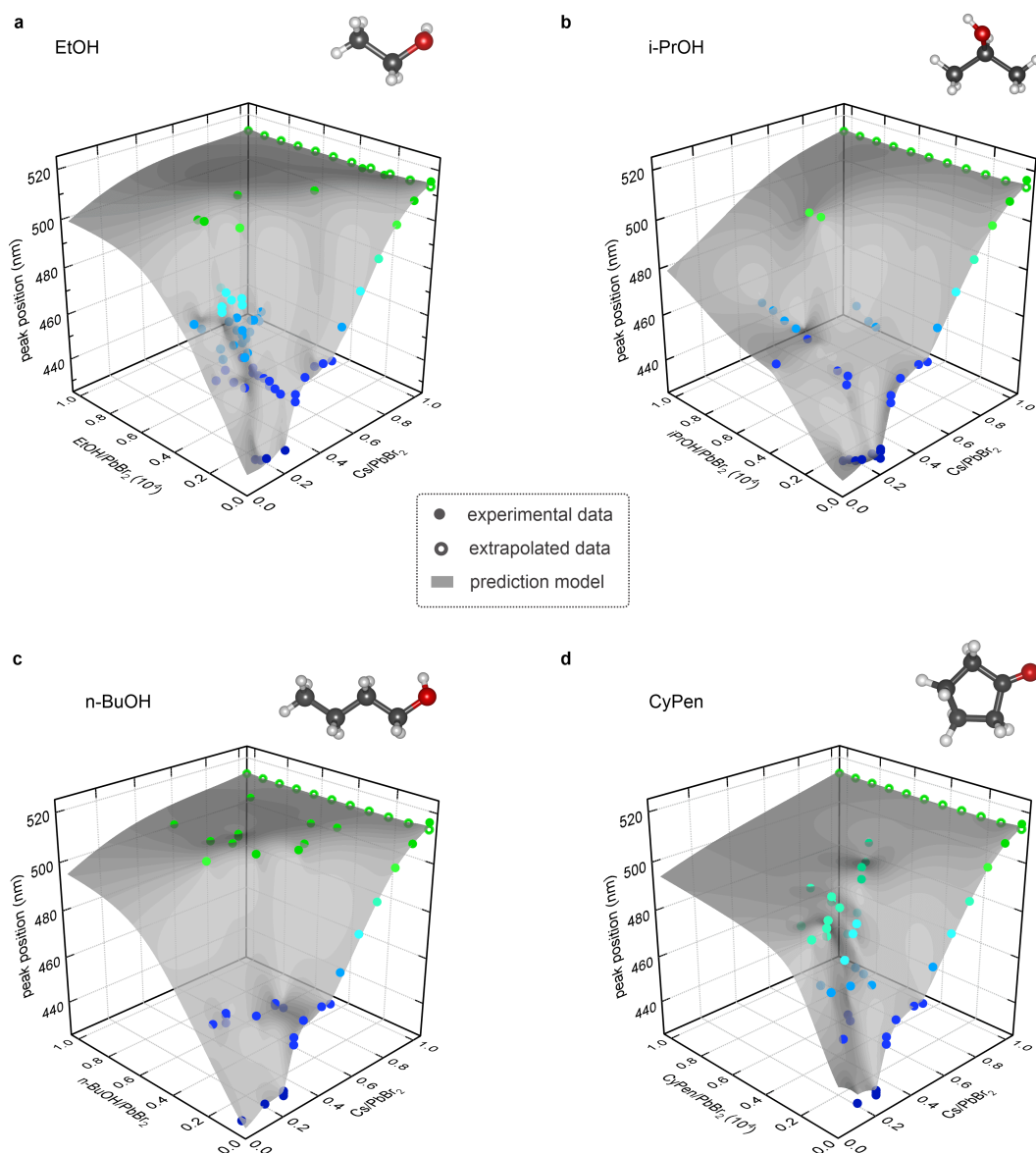
**Supplementary Table 4 | Optimized synthesis parameters for each CsPbBr<sub>3</sub> nanocrystal product.** Precursor concentrations, volumes and antisolvent type, volume as well as resulting Cs/PbBr<sub>2</sub> ratio and molar antisolvent/PbBr<sub>2</sub> ratio resulting in optimized nanocrystal products with the narrowest PL profiles. Syntheses are marked with symbols corresponding to the optimization method: *★* for direct *Synthesizer* suggestion, *c* for *confinement tuning* and *a* for optimization by *antisolvent choice*.

	V <sub>PbBr<sub>2</sub></sub> (μL)	c <sub>PbBr<sub>2</sub></sub> (M)	V <sub>Cs</sub> (μL)	c <sub>Cs</sub> (M)	antisolvent	V <sub>as</sub> (μL)	FWHM (meV)
2	2385	0.074	68	0.020	n-BuOH	2235	69 (★)
3	1000	0.010	100	0.045	-	0	73 (c)
4	2250	0.007	68	0.020	CyPen	2520	87 (★, c, a)
5	1500	0.020	150	0.020	EtOH	2000	98
6	2590	0.010	403	0.020	CyPen	5000	96 (★, a)
7	750	0.007	75	0.020	CyPen	2000	87 (a)
8	1000	0.003	100	0.020	CyPen	1335	80 (a)
∞	695	0.010	249	0.020	MeOH	1155	75

## Supplementary Figures - Prediction of photoluminescence peak wavelength

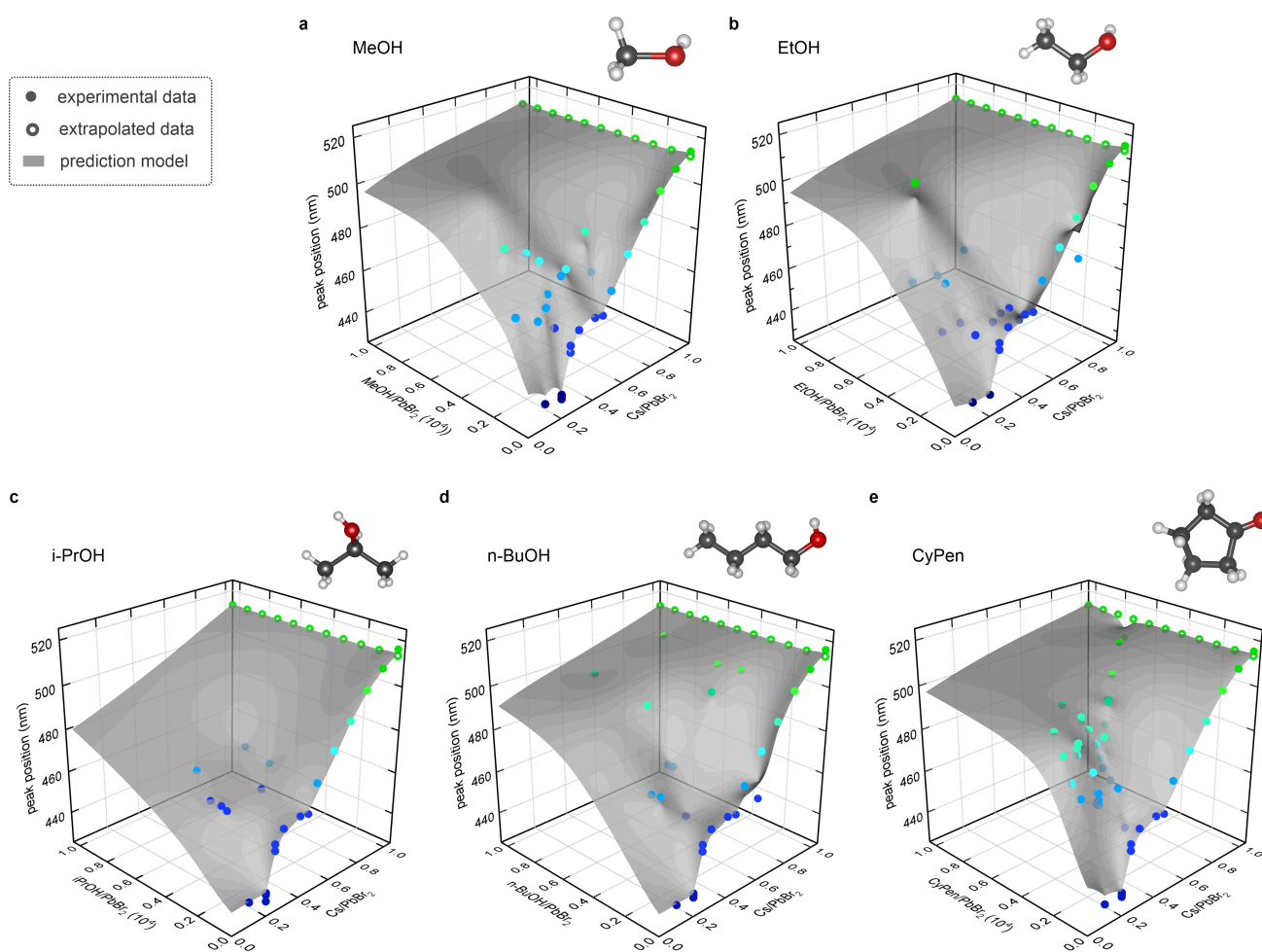


**Supplementary Fig. 1 | Correlation matrix for feature selection.** A selection of high level calculated features is compared in regard to their correlation with the PL peak position of the  $\text{CsPbBr}_3$  nanocrystal product in the **a**, precipitate (P) or **b**, supernatant (S). In general, we find that the relative molar ratios are more expressive than the individual, absolute molar values or the total volume. However, the as/Cs ratio and PL peak position show no noteworthy correlation and can be omitted. Therefore, the remaining two ratios  $\text{Cs/PbBr}_2$  and  $\text{as/PbBr}_2$  were identified as the best parameters for PL peak prediction, resulting in a reduced, two-dimensional synthesis parameter space. This systematic approach of feature selection on calculated properties gives some initial insights into the underlying chemical processes. Components whose ratios between amounts of substance correlate with the type of product are likely to interact during the synthesis.

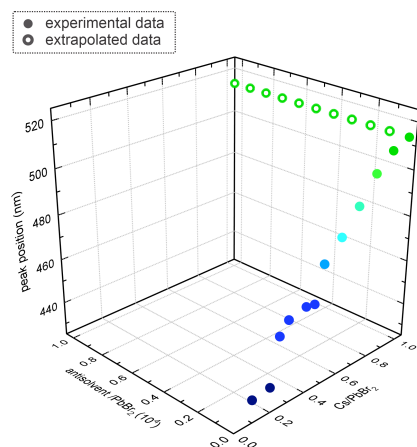


**Supplementary Fig. 2 | Three-dimensional representation of parameter space and PL peak wavelength for P-type samples synthesized with EtOH, i-PrOH, n-BuOH and CyPen antisolvent.** Two-dimensional parameter space defined by  $\text{Cs}/\text{PbBr}_2$  and antisolvent/ $\text{PbBr}_2$  ratio with corresponding true PL peak wavelengths (colored dots) and the fully trained regression model (dark grey surface), including antisolvent-independent data at antisolvent/ $\text{PbBr}_2 = 0$  and extrapolated data at  $\text{Cs}/\text{PbBr}_2 = 1$  from the baseline data frame. Overall, increasing the antisolvent/ $\text{PbBr}_2$  ratio results in a further redshifted PL peak wavelength for **a**, EtOH **b**, i-PrOH **c**, n-BuOH and **d**, CyPen. The transition from blue- to green-emitting nanocrystal products is steeper for alcohol antisolvents, especially n-BuOH, and more gradual for the ketone antisolvent, CyPen. Syntheses with i-PrOH also require a minimum  $\text{Cs}/\text{PbBr}_2$  ratio of  $\geq 0.4$  to yield green-emitting  $\text{CsPbBr}_3$  nanocrystals.

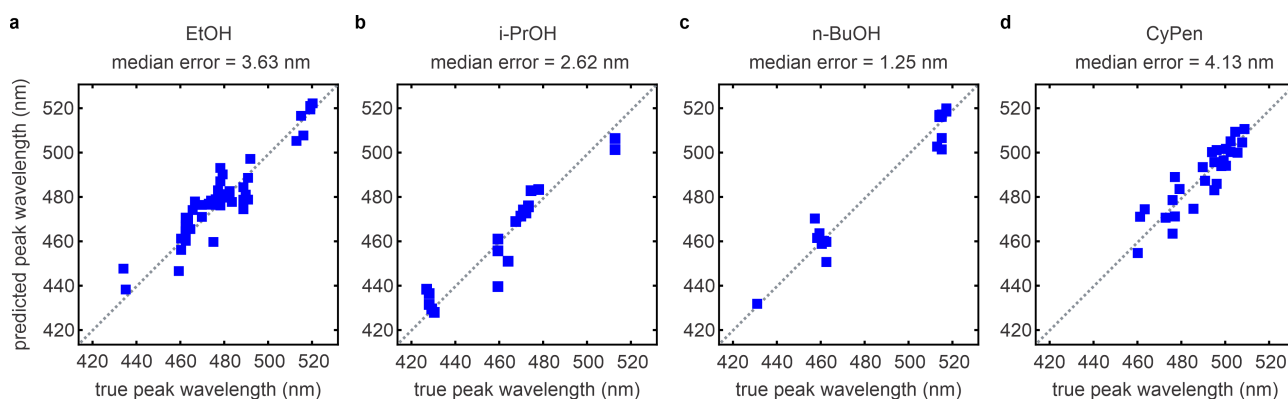




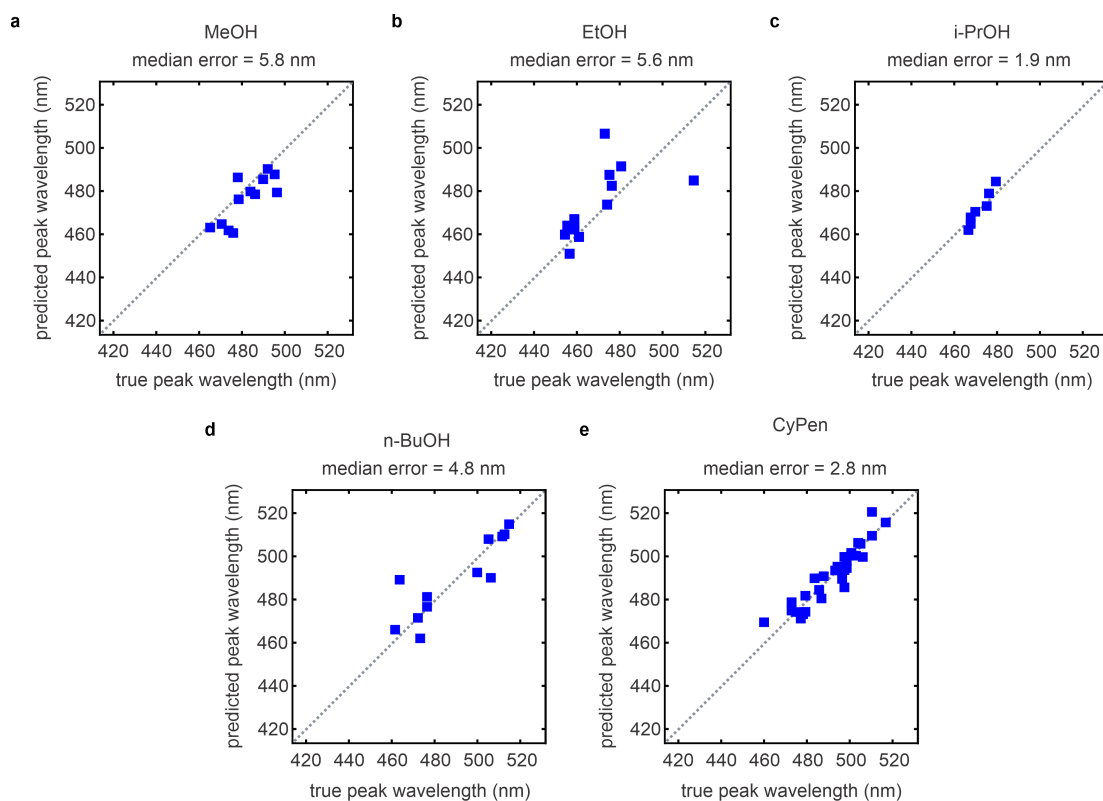
**Supplementary Fig. 3 | Three-dimensional representation of parameter space and PL peak wavelength for S-type samples.** Two-dimensional parameter space defined by Cs/PbBr<sub>2</sub> and antisolvent/PbBr<sub>2</sub> ratio with corresponding true PL peak wavelengths (colored dots) and the fully trained regression model (dark grey surface), including antisolvent-independent data at antisolvent/PbBr<sub>2</sub> = 0 and extrapolated data at Cs/PbBr<sub>2</sub> = 1 from the baseline data frame. Similarly to P-type data, increasing the antisolvent/PbBr<sub>2</sub> ratio results in a further redshifted PL peak wavelength for **a**, MeOH **b**, EtOH **c**, i-PrOH, **d**, n-BuOH and **e**, CyPen. The transition from blue- to green-emitting nanocrystal products is steeper for alcohol antisolvents, especially n-BuOH, and more gradual for the ketone antisolvent, CyPen. As seen for EtOH **b**, due to the implementation of the exponential kernel with a short length scale for the GP an isolated data point can in some rare cases lead to overfitting. However the overall validation accuracy gained from this approach far outweighs the occasional over-representation of a single sample.



**Supplementary Fig. 4 | Antisolvent independent baseline data.** This set of experimental and extrapolated data-points at the boundaries of the parameter space is inherently antisolvent independent and can therefore be added to the data set for any antisolvent. As the PL peak wavelength is assumed to be continuous, the experimental data points from syntheses with  $\text{Cs/PbBr}_2 = 0\text{-}1.0$  but without antisolvent represent the lower limit for any antisolvent molecule. The upper limit at  $\text{Cs/PbBr}_2 = 1$  is given by the bulk emission wavelength of  $\text{CsPbBr}_3$ . This approach has proven critical to confine the Gaussian Process model in low data regimes and has significantly improved the prediction accuracy. These data points have been strictly excluded from test data for LOO (Leave-One-Out) accuracy scores.

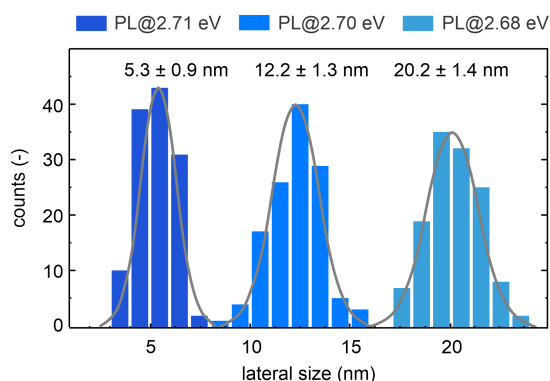


**Supplementary Fig. 5 | Prediction accuracy of photoluminescence peak wavelength for P-type data.** LOO regression plots for PL peak wavelength prediction in syntheses with different antisolvents, shown for **a**, EtOH **b**, i-PrOH **c**, n-BuOH and **d**, CyPen. A comparison between true PL peak wavelength and predicted PL peak wavelength yields low median errors ( $< \pm 4$  nm) and confirms a high prediction accuracy for all antisolvents.

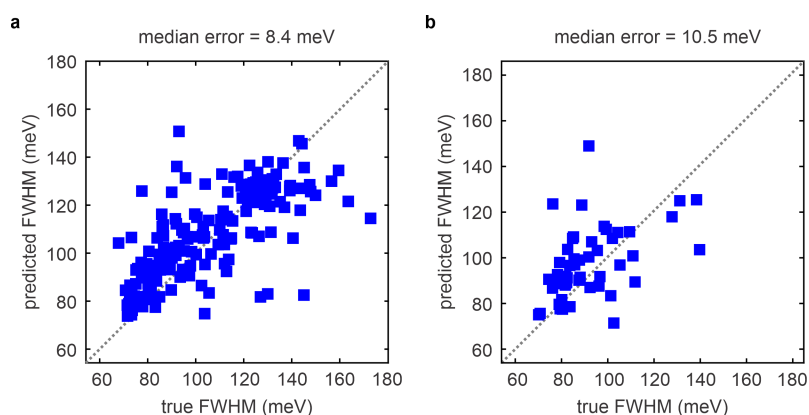


**Supplementary Fig. 6 | Prediction accuracy of photoluminescence peak wavelength for S-type data.** LOO regression plots for PL peak wavelength prediction in syntheses with different antisolvents, shown for **a**, MeOH **b**, EtOH **c**, i-PrOH, **d**, n-BuOH and **e**, CyPen. A comparison between true PL peak wavelength and predicted PL peak wavelength yields low median errors ( $< \pm 6$  nm) and confirms a high prediction accuracy for all antisolvents, similar to that for P-type samples.

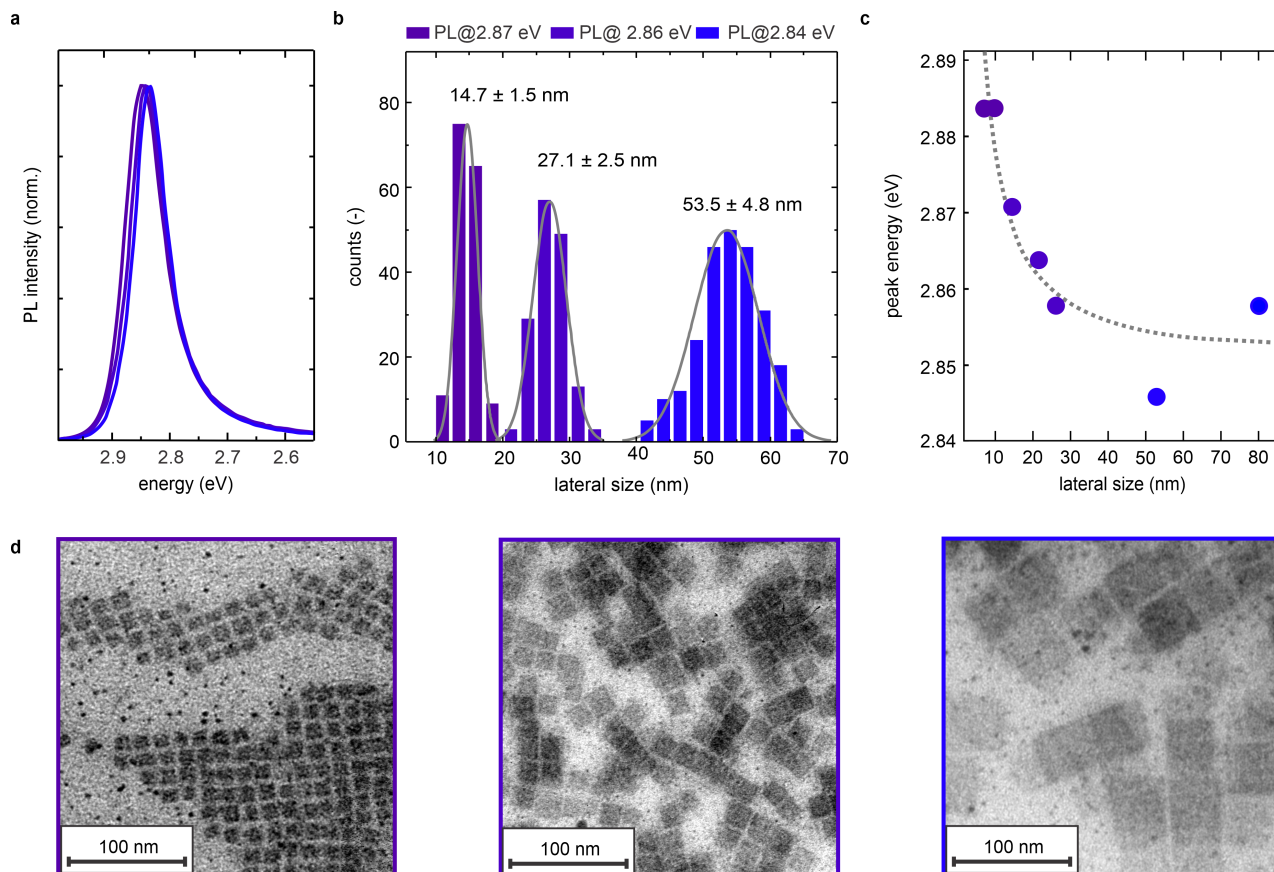
## Supplementary Figures - Optimization of full width at half maximum



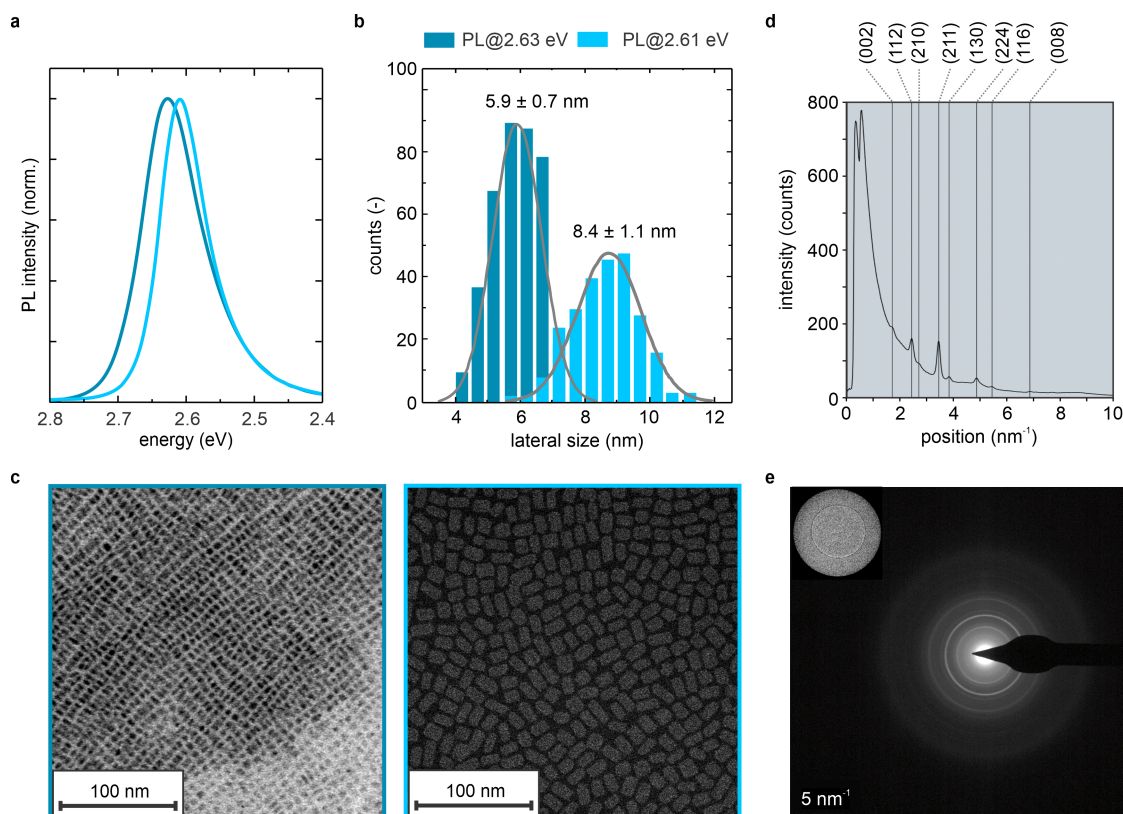
**Supplementary Fig. 7 | Lateral size distribution of 3ML CsPbBr<sub>3</sub> NPLs.** Lateral size distribution of three different 3ML NPL samples with slightly offset PL peak energies at 2.71 eV, 2.70 eV and 2.68 eV. The lateral sizes of NPLs in each sample were determined from TEM images, and amount to  $(5.3 \pm 0.9)$  nm,  $(12.2 \pm 1.3)$  and  $(20.2 \pm 1.4)$  nm.



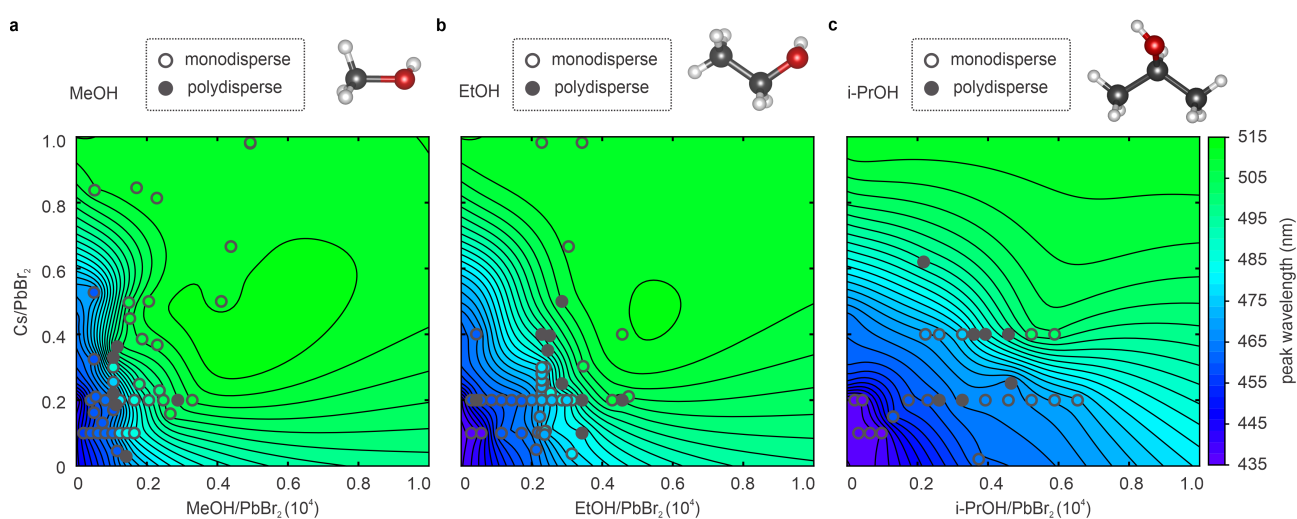
**Supplementary Fig. 8 | Prediction accuracy of full width half maximum.** LOO plots for FWHM predictions of **a**, the entire dataset with a median error of  $\pm 8.4$  meV. Due to the correlation between FWHM and peak energy, there is a chance that the model draws information about the FWHM values from an estimated peak energy as opposed to only modeling the mechanisms that lead to ideal line widths. **b**, To address this concern, samples with PL peak energy between 2.67 eV and 2.72 eV (3ML NPLs) are shown in a second LOO regression plot. The wide range of predicted FWHM values for this narrow energy interval demonstrates true predictive capabilities of the model. Yet the inherent noise observed in FWHM data might impede reliable predictions and motivates the presented hybrid approach of Gaussian Process regression combined with chemical and physical insights for FWHM optimization. For FWHM predictions, emissive and monodisperse P-type samples were used for training.



**Supplementary Fig. 9 | Lateral confinement tuning in 2ML CsPbBr<sub>3</sub> NPLs.** **a**, PL spectra and corresponding **b**, lateral size distribution of three different 2ML NPL samples with slightly offset PL peak energies at 2.87 eV, 2.86 eV and 2.84 eV. The lateral sizes amount to  $(14.7 \pm 1.5)$  nm,  $(27.1 \pm 2.5)$  nm and  $(53.5 \pm 4.8)$  nm and were determined from corresponding TEM images, shown in **d**. The correlation between NPL lateral size and PL peak energy, i.e. bandgap energy is adequately described by a power law fit in **c**. We note that scattering was observed in colloidal solutions of 2ML NPLs with lateral dimensions > 30 nm, most likely due to face-to-face stacking of the NPLs. This can result in reabsorption processes and minor shifts in the PL peak energy.

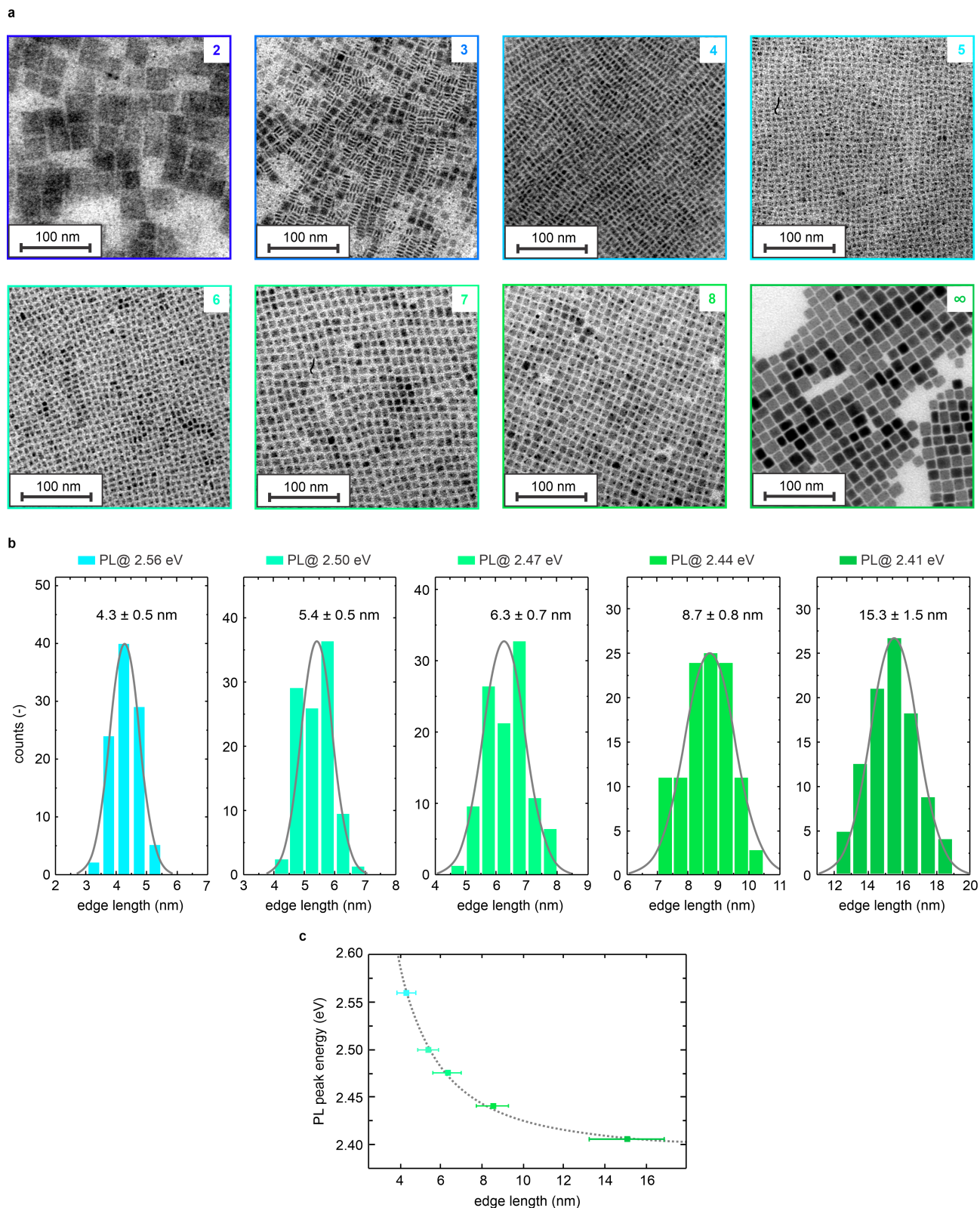


**Supplementary Fig. 10 | Lateral confinement tuning in 4ML CsPbBr<sub>3</sub> NPLs.** **a**, PL spectra and corresponding **b**, lateral size distribution of two different 4ML NPL samples with slightly offset PL peak energies at 2.63 eV and 2.61 eV. The lateral sizes amount to  $(5.9 \pm 0.7)$  nm and  $(8.4 \pm 1.1)$  nm for the smallest lateral dimensions of NPLs and were determined from corresponding **c**, TEM and annular dark field scanning transmission electron microscopy (ADF-STEM) images. **d**, Intensity profile of the diffraction pattern shown in **e**. Indices were assigned for orthorhombic CsPbBr<sub>3</sub>. ADF-STEM images of larger 4ML CsPbBr<sub>3</sub> NPLs were recorded using a probe-corrected FEI Titan Themis 60-300 operated at an acceleration voltage of 300 kV. Specimen preparation was carried out by dropcasting onto TEM grids (Quantifoil R2/2, 2 nm ultrathin carbon).



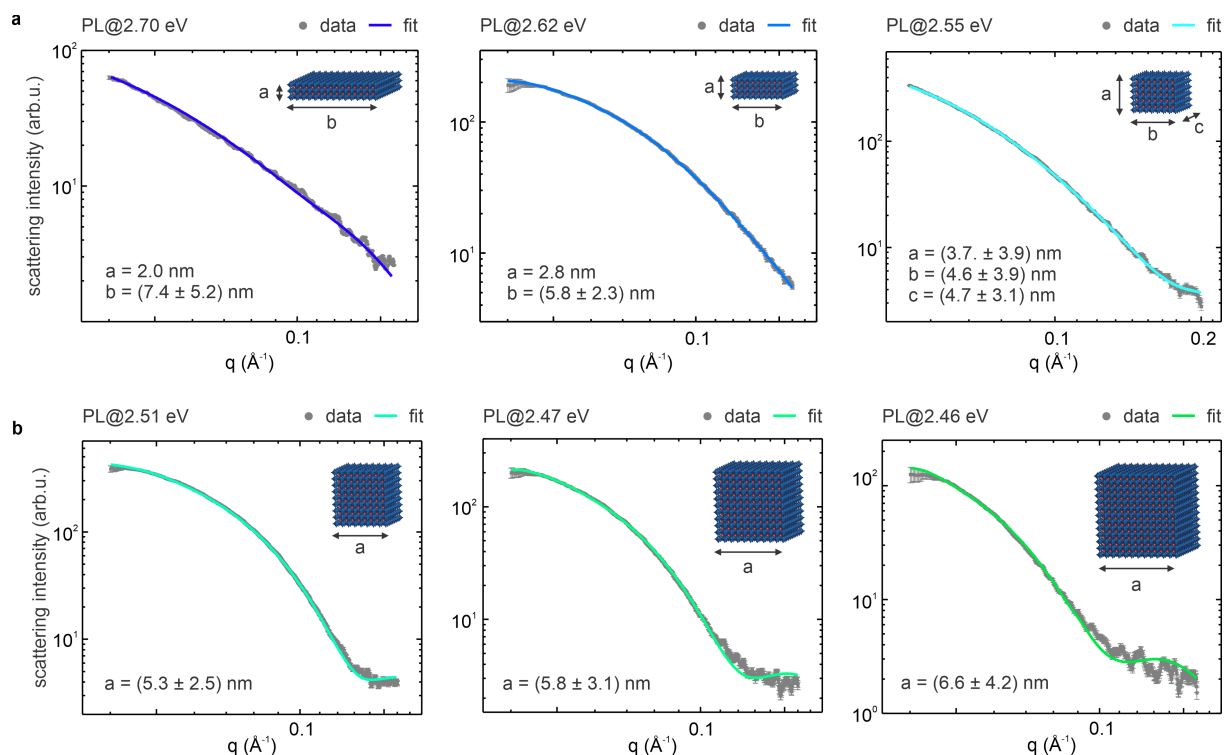
**Supplementary Fig. 11 | Two-dimensional representation of parameter space and PL peak wavelength.** Contour plots for nanocrystal syntheses with **a**, MeOH **b**, EtOH and **c**, i-PrOH, showing the characteristic behavior of the PL peak wavelength depending on Cs/PbBr<sub>2</sub> and antisolvent/PbBr<sub>2</sub> ratio as well as the distribution of monodisperse and polydisperse products in the parameter space. Polydisperse samples with FWHM > 120 meV are more frequently obtained in regions with a high PL peak wavelength gradient.





**Supplementary Fig. 12 | Morphology of CsPbBr<sub>3</sub> nanocrystal products.** **a**, Transmission electron microscopy imaging of various perovskite nanocrystal products obtained by antisolvent-assisted spontaneous crystallization in non-polar solvents. Anisotropic 2ML NPLs, 3ML NPLs and 4ML NPLs (products 2-4) as well as more isotropic, cube-shaped nanocrystals of different sizes (products 5-∞) are shown. **b**, Size distribution of products 4-7, all confirming excellent size uniformity of CsPbBr<sub>3</sub> nanocrystals. **c**, The PL peak energy of cube-shaped CsPbBr<sub>3</sub> nanocrystals is inversely correlated with the nanocrystal edge length observed in TEM images and can be accurately described with an inverse power law function.

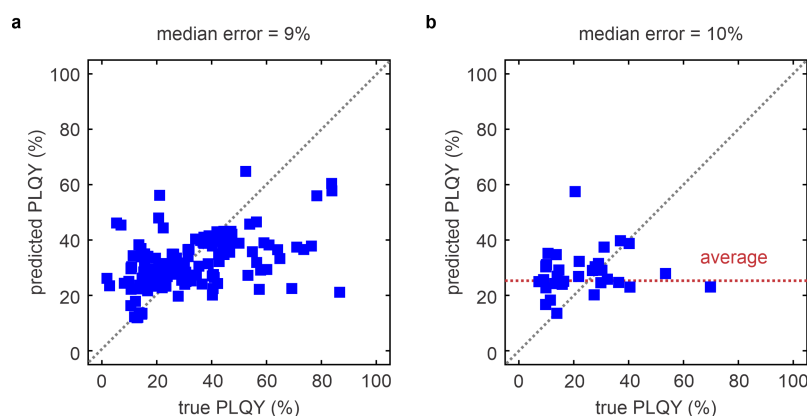
SAXS data was recorded using a Mo  $K_{\alpha}$  microfocus source ( $\lambda = 0.71 \text{ \AA}$ ) [1]. The detector was a Pilatus 100K (Dectris) at a sample-detector-distance of 0.808 m and the detector position was calibrated with silver behenate (AgBH) by using pyFAI [2]. SAXS measurements of colloidal CsPbBr<sub>3</sub> nanocrystals dispersed in n-hexane (20 mg/mL) were performed in a custom-built cell equipped with a quartz capillary. SAXS data was corrected by subtraction of the instrumental signal, air signal as well as the glass and solvent (n-hexane) signal. A python script with the module jscatter was used for fitting of the SAXS profiles [3], using the nanocrystal sizes determined from TEM images as a reference for parameter initialization. Different fit models for cuboid nanocrystals with parameters for edge lengths  $a$ ,  $b$  and  $c$  and respective standard deviations were tested. From these, only as many parameters as needed were used to obtain a good fit for the experimental data; two for a true cube (where all edge lengths are identical,  $a = b = c$ ), four for a platelet or rod (two identical edge lengths,  $a \neq b = c$ ) and six for generic cuboid ( $a \neq b \neq c$ ).



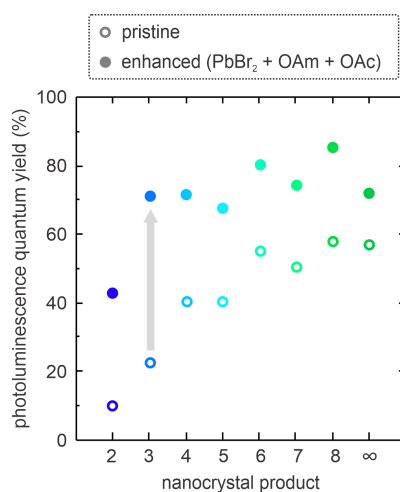
**Supplementary Fig. 13 | Small-angle X-ray scattering analysis of CsPbBr<sub>3</sub> nanocrystals measured with a laboratory setup.** **a**, The small-angle X-ray scattering (SAXS) intensity of pristine nanocrystals in n-hexane is described very well by a plate model fit ( $a < b = c$ ) for nanocrystal products 3 and 4. For nanocrystal product 5, a plate model fit of the experimental SAXS data yields two similar values for  $a$  and  $b$ , hinting at a very minor anisotropy of the perovskite nanocrystals. **b**, The SAXS intensity of nanocrystal products 6-8 is better described with a cuboid model ( $a = b = c$ ). These results are in good agreement with the sizes of similar nanocrystal samples observed in TEM images. Therefore, the SAXS analysis also confirms the different shape of nanocrystal products 2-4 (NPLs) in contrast to nanocrystal products 5- $\infty$  (nanocubes).



## Supplementary Figures - Photoluminescence quantum yield



**Supplementary Fig. 14 | Prediction accuracy of photoluminescence quantum yield.** LOO plots for PLQY predictions of **a**, the entire dataset, for which the trained model has some predictive power, and **b**, samples with PL peak energies between 2.67 eV and 2.72 eV (3ML NPLs), for which the trained model predicts an average PLQY value. This behavior is a direct consequence of the correlation between nanocrystal product and PLQY. Trained on all data points, the model accurately predicts the PL peak wavelength and estimates the PLQY based on this quantity. However, this approach fails for a narrower emission range, i.e. for a single type of NC product. Any further correlation between synthesis parameters and the observed PLQY is likely overshadowed by the inherent noise with regards to this target. For PLQY predictions, monodisperse S-type and P-type data were used for training.



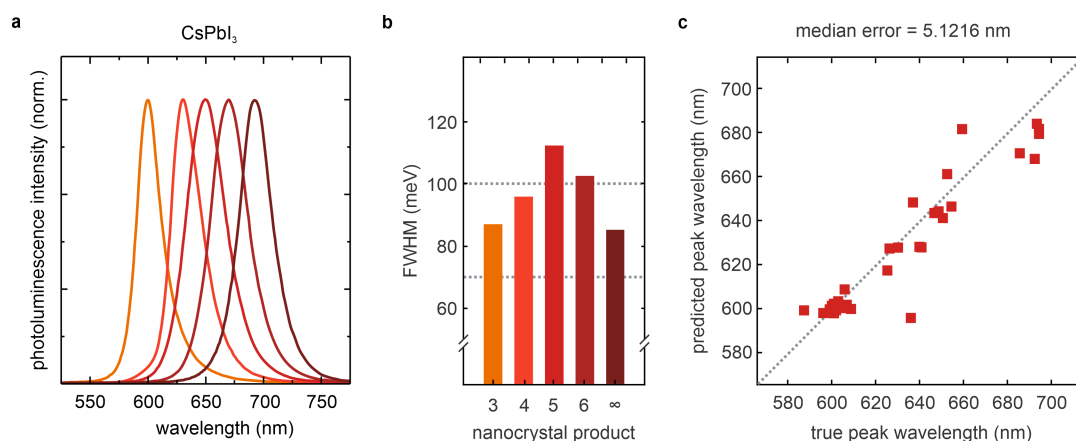
**Supplementary Fig. 15 | Photoluminescence quantum yield of pristine and enhanced CsPbBr<sub>3</sub> nanocrystals.** Comparison of PLQY values for pristine perovskite nanocrystals as obtained from antisolvent-assisted synthesis in ambient conditions and those treated with 5-10% volume of enhancement solution containing PbBr<sub>2</sub>, OAm and OAc in n-hexane. The post-synthesis treatment passivates surface vacancies and notably improves PLQY values of all CsPbBr<sub>3</sub> nanocrystal products, up to 80-90%.

## Supplementary Information - CsPbI<sub>3</sub> nanocrystals

Cs-oleate (0.2 M) was prepared by dissolving Cs<sub>2</sub>CO<sub>3</sub> (1.0 mmol, 32.6 mg) in oleic acid (10 mL) while stirring at 85 °C for up to three hours until a clear solution was obtained. Similarly, the PbI<sub>2</sub> precursor (0.01 M) was prepared by dissolving PbI<sub>2</sub> (0.1 mmol, 46.1 mg) in toluene (9.8 mL), oleic acid, and oleylamine (100 μL each) under stirring at 85 °C for up to three hours until a clear solution was obtained.

Syntheses were carried out in ambient conditions (air, 20-40% humidity and T = 20-25 °C). A glass vial was charged with PbI<sub>2</sub> precursor and Cs-oleate was injected while stirring vigorously. The reaction mixture was stirred until perovskite nanocrystal precipitation occurred and then centrifuged (3 minutes, 15856 g). The supernatant (S) was decanted and the red precipitate (P), which contains CsPbI<sub>3</sub> nanocrystals, was redispersed in 1 mL n-hexane and kept for further characterization. Insoluble byproducts in redispersed samples were removed in a second centrifugation step (3 minutes, 2097 g), if necessary.

Optical characterization was performed immediately after synthesis and purification. PL spectra of diluted CsPbI<sub>3</sub> nanocrystals were measured with a commercial FluoroMax-4Plus spectrometer equipped with a xenon arc lamp. The excitation wavelength was set to 480 nm and spectra were recorded in the range between 500-800 nm.



**Supplementary Fig. 16 | Application of *Synthesizer* to synthesis of CsPbI<sub>3</sub> nanocrystals.** **a**, Photoluminescence spectra of different isolated CsPbI<sub>3</sub> nanocrystal products with emission at 598 nm, 630 nm, 650 nm, 670 nm and 692 nm. **b**, LOO regression plots for PL peak wavelength prediction for CsPbI<sub>3</sub>. Nanocrystals were synthesized following the approach of ligand-assisted spontaneous crystallization in nonpolar solvents and ambient conditions. Similarly to the CsPbBr<sub>3</sub> system, variation of the Cs/PbI<sub>2</sub> precursor ratio enables the synthesis of CsPbI<sub>3</sub> nanocrystals with tunable orange to deep-red emission, covering the range between 595-695 nm. The injection of polar antisolvents was omitted to avoid phase transformation to non-emissive orthorhombic  $\delta$ -CsPbI<sub>3</sub> [4]. Even for a limited data set, a comparison between true and predicted PL peak wavelength yields low median errors < ±5 nm and confirms a good prediction accuracy by the *Synthesizer*. **c**, CsPbI<sub>3</sub> nanocrystal products exhibit narrow PL peaks, with FWHM values between 85-115 meV.

## References

- [1] L. K. Bruetzel et al. A Mo-anode-based in-house source for small-angle X-ray scattering measurements of biological macromolecules. *Rev. Sci. Instrum.* 87.2 (2016). DOI: [10.1063/1.4940936](https://doi.org/10.1063/1.4940936).
- [2] J. Kieffer and D. Karkoulis. PyFAI, a versatile library for azimuthal regrouping. *Journal of Physics: Conference Series* 425.20 (2013), p. 202012. DOI: [10.1088/1742-6596/425/20/202012](https://doi.org/10.1088/1742-6596/425/20/202012).
- [3] R. Biehl. Jscatter, a program for evaluation and analysis of experimental data. *PLOS ONE* 14.6 (2019), e0218789. DOI: [10.1371/journal.pone.0218789](https://doi.org/10.1371/journal.pone.0218789).
- [4] J. Sun et al. Polar Solvent Induced Lattice Distortion of Cubic CsPbI<sub>3</sub> Nanocubes and Hierarchical Self-Assembly into Orthorhombic Single-Crystalline Nanowires. *Journal of the American Chemical Society* 140.37 (2018), 11705–11715. DOI: [10.1021/jacs.8b05949](https://doi.org/10.1021/jacs.8b05949).